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DISTRIBUTION OF THE LATTICE ENERGY IN CUBIC CRYSTALS AND ITS VARIATION WITH COMPRESSION OR EXPANSION*

M. DRECHSLER

Fritz-Haber-Institut der Max-Planck-Gesellschaft, Berlin-Dahlem, Germany

and

J. F. NICHOLAS

Division of Tribophysics, CSIRO, University of Melbourne, Australia

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Abstract—The total lattice energy of face-centred and body-centred cubic crystals has been calculated on the assumption of a pair-wise interaction potential, for a variety of such potentials. These were either of Morse or Mie (Lennard–Jones) type, with the parameters adjusted to minimize the total lattice energy for an ideal unstrained crystal at 0°K. The same potentials were used to compute the variation in lattice energy when the lattice is allowed to expand or contract by up to 2.5 per cent. The relative contributions of different interactions to the total energy are given. For face-centred cubic crystals, most potentials predict a major contribution from nearest-neighbour interactions, although the contributions of second and higher neighbours are still significant. For body-centred cubic crystals, the dominant contribution usually comes from the second neighbour interactions. Some typical variations in these contributions with changes in lattice parameter are given.

1. INTRODUCTION

IN THIS paper, we assume that the total lattice energy (cohesive energy) of a crystal can be calculated as the sum of pair-wise interactions between all atoms. Various forms are considered for the interaction potential function, in all of which the parameters have been adjusted to minimize the lattice energy for an ideal unstrained crystal at 0°K. The lattice energy is then calculated for body-centred and face-centred cubic crystals, first for a lattice parameter corresponding to this minimum energy and then also for parameters deviating from this value by a few per cent. Such changes in lattice energy may be expected if the crystal is at higher temperatures or under hydrostatic pressure. Clearly any vibrational contribution to the lattice energy is not taken into account in this calculation. Corresponding calculations of

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surface energy are discussed in the following paper.⁽¹⁾

The calculation of energies by summing pairwise interactions has been used previously by many authors,^(2,3) despite the recognition that many-body interactions may make a significant contribution to the total energy.⁽⁴⁾ The attractions of the method are that it is mathematically tractable and that it does appear to provide a good approximation in many cases. The pair-wise potential itself can be chosen either as a "real" potential derived from quantum-mechanical calculations or as a "fictitious" potential⁽²⁾ of simple mathematical form and containing one or more adjustable parameters. The second approach is more in keeping with the philosophy of the method and may be expected to give a better approximation when the parameters are chosen to fit some specific experimental data.

As discussed in detail in Section 2, we have restricted our calculations to Morse and Mie (Lennard-Jones) potentials without any truncation represents the contribution to the lattice energy of the potential. Truncation, which corresponds to considering only, say, the first three or four shells of neighbours, involves an extra assumption that is unnecessary when high-speed computers are available and highly undesirable when more than one crystal structure is being considered for the same potential. Thus our calculations give precise results for ideal crystals interacting with potentials of the precise form given in equations (2) or (3).

We considered using also oscillatory potentials such as have been discussed recently, (5-8) but did not pursue this investigation for the following reasons:

(i) The only explicit potentials available are those of JOHNSON, HUTCHINSON and MARCH,⁽⁷⁾ and since these involve a number of parameters derived specifically for each metal, generalizations become very difficult.

(ii) In some associated calculations on surface energy,⁽⁹⁾ out of the seven potentials available for cubic metals, one was divergent, one gave negative values of surface energy and a third gave a degree of anisotropy of more than 3, which is very much greater than that found by any other calculation or experiment.

2. METHOD OF CALCULATION

If the interaction energy of two atoms at a distance, r, is given by E(r), then, for a perfect infinite crystal, the total lattice energy per atom is given by

$$E_G = \frac{1}{2} \sum_i M_i E(r_i),$$

(1)

(2)

where r_i is an interatomic distance, M_i is the number of neighbours of a given atom at this distance, and the sum extends over all possible r_i . The method of calculation is therefore simply to list for each structure the possible values of r_i and their associated multiplicities Mi, and then evaluate the lattice sum for particular potential functions E(r). This corresponds to rewriting equation (1) in the form

$$E_G = \sum_i E_i^S ,$$

where

$$E_i^{S} = \frac{1}{2}M_i E(r_i)$$

from the *i*-th shell of neighbours.

The potentials considered were either Morse potentials of the form

 $E(r)/E_0 = [1 - \exp\{-a(r - r_0)/r_0\}]^2 - 1,$ (4)

or Mie (Lennard-Jones) potentials of the form

 $E(r)/E_{0} = \{n(r_{0}/r)^{m} - m(r_{0}/r)^{n}\}/(m-n), (5)$

where, in all potentials, E_0 corresponds to the maximum interaction energy which occurs at a separation r_0 .

Calculations were made for various values of the Morse constant a between 3 and 6, while for the Mie functions the attractive exponent m varied from 4.5 to 9 and the repulsive exponent n(always greater than m) ranged from 6 to 14. The scaling factors E_0 were left arbitrary since only relative energy values are considered but the r_0 values were adjusted, as described below, to produce a minimum E_G for given a or (m, n). Figure 1 shows some Morse functions, while some typical Mie functions are plotted in the following paper (1) in comparison with a Morse function with a = 4.

All summations with Morse potentials were carried through for the first 300 different values of r_i , i.e. for the first 300 shells of neighbours. This corresponds to considering all interactions out to 17-18 times r_1 , and ensured that all energy values were correct to at least 8 significant figures. For Mie potentials, the initial summations were carried through for 500 shells of neighbours and the effects of more distant interactions taken into account by means of an integral approximation so that the overall accuracy should be comparable to that for the Morse potentials.

(a) Relation between the lattice parameter and r_0

From equations (4) and (5) it can be seen that distances enter only through the dimensionless variable (r/r_0) , while distances in the crystal are available in terms of the lattice parameter or of the nearest-neighbour distance, r_1 . Thus, in order to calculate the sums, the ratio r_1/r_0 must be specified so that we can write

$$r/r_0 = (r/r_1) \cdot (r_1/r_0).$$

(6)

Then, for the interatomic distances r_i , the ratios $b_i = r_i/r_1$ are fixed and known for a particular (3)

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FIG. 1. Morse functions for various values of the Morse constant a.

 $b_1 = 1, b_2 = 2/\sqrt{3}, b_3 = \sqrt{(8/3)}$, etc., and r/r_0 is are also given in Table 1. now determinate.

In this notation, the lattice energy with a Mie potential is

$$E_{G} = \frac{E_{0}}{2} \{n(r_{0}/r_{1})^{m}S(m) - m(r_{0}/r_{1})^{n}S(n)\}/(m-n),$$
(7)

where

$$S(j) = \sum_{i} M_i b_i^{-j}.$$
 (1)

Then, minimum E_G is easily shown to occur for $r_1 = r_1^*$, where

> $(r_0/r_1^*) = \{S(m)/S(n)\}^{1/(n-m)}.$ (9)

The lattice sums S(i) have been realculated by JONES and INGHAM⁽¹⁰⁾ for all integral i from 4 to 30 and appear in their Table 1 as B_i for the bodycentred cubic and C_i for the face-centred cubic lattice. Since we wished to use some half-integral exponents we have recalculated S(j) for all integral and half-integral values from 3.5 to 20. Values of (r_0/r_1^*) are given in Table 1 for various values of (m, n).

For the Morse potentials, a similar procedure can be followed but (r_0/r_1^*) cannot be evaluated in terms of simple lattice sums and a numerical

lattice, e.g. for the body-centred cubic lattice, (r_0/r_1^*) appropriate to various Morse constants a

It may be noted here that GIRIFALCO and WEIZER⁽³⁾ have considered a range of materials and used experimental values of sublimation energy, lattice parameter, and compressibility to deduce the values of E_0 , a, and r_0 appropriate to each. Table 2 lists these as some of our calculations have been carried through with them. Although they express the Morse potential in a different form their values correspond to the same minimization criterion that we have used. Other workers have used a similar approach to deduce the (m, n) values of Mie potentials appropriate to different materials.⁽²⁾ Some discussion of this appears in Section 4(b) of the following paper.⁽¹⁾

(b) Variation of the lattice parameter

In studying the effect of temperature or pressure on the lattice energy, we assume that the nearest-neighbour distance in a perfect unstrained crystal at 0°K is r_1^* and that the only effect of temperature or pressure is to vary the r_1 by a few per cent from this value. The sums such as those in equation (5) can then be evaluated to show the variation in lattice energy with lattice parameter. This has been carried out for r_1/r_1^* deviating from unity by up to ± 2.5 per cent for the Mie potentials and up to ± 5 per cent for the Morse potentials.

In order that the variation in E_G , expressed here minimization procedure is necessary. The values of in terms of a variation in lattice parameter, can be

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Table 1. Ratio of the nearest-neighbour distance, r1*, for minimum lattice energy to the distance of the minimum in the potential function, r_0 . For ratios of lattice parameter to r_0 , the values given must be multiplied by $\sqrt{2}$ for face-centred cubic and $2/\sqrt{3}$ for body-centred cubic crystals

Face-centred cubic							
Morse $a r_1 * / r_0$	=	3 0.6346	3.5 0.8266	4 0·9014	4·5 0·9395	5 0·9612	6 0·9825
Mie m	=	4.5	5	6	7	8	9
n = 6		0.8140	0.8519				
7		0.8564	0.8873	0.9243			
8		0.8844	0.9104	0.9411	0.9583		
10		0.9182	0.9379	0.9607	0.9731	0.9806	0.9855
12		0.9375	0.9532	0.9712	0.9809	0.9866	0.9903
14		0.9497	0.9628	0.9776	0.9855	0.9901	0.9930

3·5 0·8035	4	4.5	E	
	0.8767	0.9147	0.9367	6 0.9601
5	6	7	8	9
32 0·8303	0.0001			
29 0.8886	0.9193	0.9368		
74 0.9169 78 0.9335	0.9399	0.9529 0.9621	0.9611 0.9686	0.9666
) ;; ;)	5 0.32 0.8303 0.8655 0.9 0.8886 0.9169 78 0.9335 12 0.9444	5 6 32 0.8303 51 0.8655 0.9021 529 0.8886 0.9193 774 0.9169 0.9399 78 0.9335 0.9519 112 0.9444 0.9597	5 6 7 32 0.8303	$\begin{array}{cccccccccccccccccccccccccccccccccccc$

Table 3. Temperatures and pressure necessary to change lattice parameters by particular amounts(11,12)

Material	Temperature the lattice para 0°K va	(°K) at which ameter exceeds alue by	Pressure (10 ³ kg/cm ²) which changes the lattice parameter at room temperature by			
	1 per cent	2 per cent	-0.5 per cent	-1 per cent	-2 per cent	
Na	180	350	1.0	2.0	4.2	
Al	. 450	800	11.5	25	57	
Cu	650	1250	23.5	53	~200	
W	1950	3250	70		-00	

considered in terms of temperature or pressure, expected for 1 per cent and 2 per cent change in Table 3 gives the relevant experimental data for a parameter. few metals.

3. RESULTS

Since E_G has been minimized at r_1^* , the changes in the total lattice energy are all small but much greater changes occur in the energies associated Table 4 shows the calculated lattice energy for with the interactions of particular atoms or groups various potentials together with the variations of atoms. Thus, Fig. 2 shows, for a Morse a = 4



 $\begin{array}{c} 0.04644\\ 3.098\\ 0.6743\end{array}$

0-04485 3-142 0-6947

 $\begin{array}{c} 0.06334 \\ 3.148 \\ 0.6972 \end{array}$

 $\begin{array}{c} 0.05424 \\ 3.170 \\ 0.7065 \end{array}$

 $\begin{array}{c} 0.1416\\ 3.530\\ 0.8096 \end{array}$

0.4174 3.951 0.8716

0-9906 4-279 0-9006

0-4414 4-330 0-9041

0-8032 4-488 0-9140

 $\begin{array}{l} E_0(=D), \text{ in eV} \\ a \ (=\ln\beta) \\ r_1^* / r_0 \ (=\sqrt{3} \alpha a_0 / \ln\beta) \end{array}$

Rb

S

Na

M

Ba

Fe

M

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Mo

Metal

Table 2. Constants in the Morse potential as deduced by GIRIFALCO and WEIZER.⁽³⁾ The expressions in brackets use the symbols from (3) 0-1513 3-680 0-8602 Sr $\begin{array}{c} 0.1623\\ 3.680\\ 0.8602 \end{array}$ Face-centred cubic crystals Ca Body-centred cubic crystals 0-2703 3-788 0-8762 Al 0-3429 3-894 0-8898 Cu 0-4205 3-947 0-8958 Z 0-3323 4-265 0-9244 Ag 0-2348 4-419 0-9348 Pb $\begin{array}{l} E_0(=D), \text{ in eV} \\ a(=\ln\beta) \\ r_1^*/r_0 (=\sqrt{2\alpha a_0}/\ln\beta) \end{array}$ Metal

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FIG. 2. Interaction energy (full line) between, and force exerted on (broken line) first to fourth nearest neighbours in a body-centred cubic crystal with a Morse a = 4potential. The changes caused by a 2 per cent lattice expansion or contraction are also shown.

Table 4. Lattice energies, E_{α} for different potentials and the differences caused by small variations of the lattice parameter from its value for minimum E_G . Since E_G is itself negative, the quoted increases are in fact decreases in the absolute value of E_G

			Per cent increase in E_{α} for lattice				
	$-E_{g}/E_{0}$		expansion of		contraction of		
Potential	f.c.c.	b.c.c.	1 per cent	2 per cent	1 per cent	2 per cent	
Morse: $a =$							
3	22.802	22.806	0.04	0.17	0.05	0.19	
3.25	15.895	15.881	0.07	0.27	0.07	0.30	
3.5	12.819	12.788	0.09	0.35	0.10	0.40	
3.75	11.049	11.000	0-11	0.44	0.12	0.50	
4	9.900	9.830	0.13	0.52	0.15	0.60	
4.25	9.097	9.007	0.16	0.60	0.17	0.71	
4.5	8.509	8.397	0.18	0.69	0.20	0.82	
4.75	8.064	7.927	0.20	0.78	0.22	0.93	
5	7.718	7.555	0.23	0.87	0.25	1.05	
5-25	7.443	7.253	0.25	0.96	0.28	1.18	
5.5	7.222	7.003	0.28	1.06	0.31	1.31	
5.75	7.041	6.793	0.31	1.15	0.34	1.45	
6	6.892	6.612	0.33	1.26	0.38	1.59	
Mie: $(m, n) =$							
(4.5, 6)	24.840	24.609	0.13	0.49	0.14	0.59	
(4.5, 14)	12.410	11.951	0.29	1.10	0.34	1.46	
(5,7)	15-423	15.198	0.17	0.63	0.18	0.77	
(5, 8)	13-567	13.319	0.19	0.72	0.21	0.89	
(6, 12)	8.610	8.237	0.34	1.26	0.39	1.66	
(9, 14)	6.656	6.089	0.58	2.13	0.69	3.01	

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potential in a body-centred cubic crystal, how the with $a \leq 3.1$. Thus even for quite large comenergies of interaction between first to fourth pressions, second-neighbour interactions should nearest-neighbours are related for $r_1/r_1^* = 0.98$, remain attractive. In terms of energy contributions, 1.0, and 1.02. The derivative of this energy, which Fig. 4(a) shows that a nearest-neighbour interacgives the force on one atom exerted by the appropriate neighbours, is also plotted here. More a > 3.8 and this effect is reinforced in the condetails of these energies are given in Fig. 3 for a variety of Morse constants, while Fig. 4 shows the Fig. 5(a)]. For Mie potentials, calculations show variation with parameter of $E(r_1)/E(r_2)$. Figure 5 shows E_i^s (i = 1, ..., 4) as functions of the that $E_1^s/E_2^s > 1$ whenever 2m+n > 16. parameter for two different Morse constants. In body-centred cubic crystals, a very different

4. DISCUSSION OF THE RESULTS

Although the variation of E_G with lattice parafollowing generalizations are of interest.

in parameter and given potential is effectively the same for each structure. This can be shown to be 5 show that, with potentials appropriate to ideal rigorously true for Mie functions and the maximum variation for Morse functions is 0.003 per nearest-neighbour interaction for a > 5.1 or cent, at a = 6.

the potential functions, the changes for compression exceed those for expansion.

of E_{G} , the Mie potentials give a greater variation 5.1 is sensitive to such changes, decreasing to with parameter than do the Morse potentials.

For particular interactions, it is clear on physical grounds that minimizing the lattice energy must most parameter values, the largest energy is in fact lead to an r_1^* less than r_0 , i.e. to a repulsive force between nearest-neighbours in the unstrained This of course follows also from Table 1 since crystal. Any compression obviously increases this model for most potentials, the exceptions being body-centred cubic crystals, the region with those few for which $r_1 * / r_0 > 0.98$.

For second-neighbour interactions, whose significance has been discussed in some detail by attractive so long as

0.7071 for face-centred cubic crystals, $r_1/r_0 > 1/b_2 =$ 0.8660 for body-centred cubic crystals.

For face-centred cubic crystals, Table 1 shows that this limit is attained only for Morse potentials theory based on nearest-neighbour interactions

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tion exceeds that of a second neighbour for tributions to lattice energy since $M_1/M_2 = 2$ [see that $E(r_1)/E(r_2) > 1$ whenever 2m+n > 17 and

situation prevails. Table 1 shows that for Morse potentials with a < 4, the second-neighbour interactions will be repulsive and this is also true for some of the Mie potentials. Since meter, as shown in Table 4, is in fact small, the third neighbours become repulsive only for $r_1/r_0 < 1/b_3 = 0.6124$, interactions from these (i) The fractional change in E_G for a given change and further neighbours should be attractive. In terms of energy, calculations of $E(r_i)$ for i = 1, ...,crystals, the largest energy is associated with the 3m+n > 29, with the second-neighbour interac-(ii) As is to be expected from the asymmetry of tion for $5 \cdot 1 > a > 3 \cdot 3$ or 3m + n < 29, and with the third neighbour interaction for a < 3.3. The limit of 3.3 is insensitive to changes in parameter (iii) For potentials producing comparable values but, as can be seen from Fig. 4(b), the limit at about 4.6 for 2.5 per cent lattice expansion. In summary, it appears that for most potentials and associated with a second-neighbour interaction. r_1/r_0 near 0.87 implies that the second neighbours repulsion while Table 1 shows that, after 2 per are separated by about r_0 , where the interaction cent expansion, the force is still repulsive on this energy approaches its maximum value of E_0 . For a < 3.5 which may be appropriate for the alkali metals (Table 2) shows a further interesting feature. For these relatively flat potentials (see Fig. 1), the DRECHSLER and LIEPACK,⁽¹³⁾ the force will be nearest-neighbour interaction becomes of opposite sign to the other interactions [Fig. 4(b)] and may be of considerable magnitude. At this stage, as pointed out above, the effect of third neighbours is very significant, particularly in contributions to total lattice energy since $M_1: M_2: M_3 = 8:6:12.$

The considerations above point out the considerable errors which can be introduced by a 2604

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FIG. 3D FIG. 3. Interaction energy as a function of lattice parameter for various neighbours interacting with Morse potential of (a) a = 4 and 5.5, in a face-centred cubic crystal, and (b) a = 3.5, 4, and 5, in a body-centred cubic crystal. The broken lines show a fraction of the total lattice energy.

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FIG. 4. Variation with lattice parameter of $E(r_1)/E(r_2)$ for various Morse potentials in (a) a face-centred cubic crystal and (b) a body-centred cubic crystal.







FIG. 5b

FIG. 5. Variations with lattice parameter of various contributions to E_0 in (a) a face-centred cubic crystal for a Morse potential of a = 3.89 and (b) a body-centred cubic crystal for a Morse potential of a = 4.28; according to (3) these values are appropriate to Cu and W. The number of atoms in each shell of neighbours is given in brackets.

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only, particularly for body-centred cubic crystals. One direct example of this has been pointed out by DRECHSLER and LIEPACK⁽¹³⁾ in connection with the growth of a (110) face in a body-centred cubic crystal. Here, a single adatom on top of such a face could locate itself on a site with three nearestneighbours but instead, since growth occurs on such faces, must be located on a site with two nearest and two second-nearest neighbours. This is consistent with detailed calculations with pairwise potentials which show that the latter site has a lower energy.

An interesting demonstration of the effects of distant neighbours has been produced by constructing ball-and-spring models of body-centred and face-centred cubic crystals.(14) Conventional models of this type use springs only between nearest neighbours and since these are made identical they are all in equilibrium for an unstrained crystal. However, the new models use springs between atoms up to third-neighbour separation with spring constants adjusted to fit particular interaction potentials. In these, of course, the nearest-neighbour springs are always in compression and the model gives notably different results from the conventional one when used to study the stability of structures, their elastic properties or the positions of surface atoms relative to their ideal lattice positions.

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